

32–292 K is 0.95, which gives  $S = 1.46$ , which is higher than the expected value of 1.1 for 2.2 holes per formula in  $Tl_{1.8}Re_6Se_{12}$ . The reason for this is not yet clear.

The resistivity and magnetic measurement results are as expected. Each rhenium atom is five-coordinated by selenium atoms in a square-pyramidal geometry ( $dsp^3$ ), which leaves four d orbitals for each rhenium atom. According to molecular orbital calculations for  $M_6X_8$  clusters,<sup>18–20</sup> there are 24 orbitals per cluster, 12 of which are bonding and 12 are antibonding orbitals with a substantial energy gap between. In  $Tl_4Re_6X_{12}$ , Tl and Re ions are formally in +1 and +3 oxidation states, respectively,<sup>21</sup> while both X and  $X_2$  are in –2 formal oxidation state. There are 24 electrons per cluster unit, which completely fill the bonding states and leave the antibonding states empty. This explains the semiconducting behavior. The magnitude of the energy gap between bonding and antibonding states is determined by the nature of the bridging groups. Since Se 4p orbitals have greater overlapping ability than S 3p orbitals, it is not surprising that the resistivity and the activation energy of the sulfide are much higher than

corresponding values of the Se analogue. Since all the electrons are paired, the diamagnetic behavior of  $Tl_4Re_6Se_{12}$  seen in Figure 4 is not unexpected. The upturn of the susceptibility at low temperature found for  $Tl_4Re_6Se_{12}$  is probably due to the existence of paramagnetic impurities in the sample. In  $Tl_{1.8}Re_6Se_{12}$ , holes are expected in the valence band and metallic and Pauli paramagnetic behavior would be anticipated. The source of the observed paramagnetic behavior (Figure 4) is not understood at this time. The magnetic data are consistent with the p-type behavior indicated by the room-temperature Seebeck measurement.

**Acknowledgment.** We are indebted to Dr. Tom Halbert of Exxon Research and Engineering Co., Annandale, NJ 08801, for helping us with X-ray data collection and structure solution. This research was partially supported by the Office of Naval Research, the National Science Foundation—Solid State Program, under Grants DMR-84-04003, DMR-84-08266, DMR-87-14072, and DMR-87-05620.

**Registry No.**  $Tl_4Re_6Se_{12}$ , 117201-42-0;  $TlSe$ , 12039-52-0; Re, 7440-15-5; Se, 7782-49-2;  $Tl_2Se$ , 15572-25-5;  $Tl_4Re_6S_{12}$ , 117201-37-3;  $TlS$ , 12039-09-7;  $Tl$ , 7440-28-0; S, 7704-34-9.

**Supplementary Material Available:** For  $Tl_4Re_6Se_{12}$ , Tables SI, SIII, and SIV, listing crystal and refinement data, anisotropic thermal parameters, and nonessential bond distances and angles (7 pages); Table SII, listing observed and calculated structure factors (5 pages). Ordering information is given on any current masthead page.

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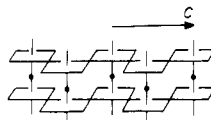
## Resistivity Anomalies of the Diphosphate Tungsten Bronze $Cs_{1-x}P_8W_8O_{40}$ ( $x = 0-0.46$ ) and Its Partially Substituted Phases $Cs_xA_yP_8W_8O_{40}$ ( $A = Rb, Na$ ) and $CsP_8W_{8-x}Mo_xO_{40}$ : Synthesis, Physical Property Measurements, and Band Electronic Structure Calculations

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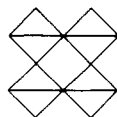
Received November 8, 1988

The cesium phosphate tungsten bronze  $CsP_8W_8O_{40}$  exhibits a resistivity hump at  $\sim 160$  K and a resistivity upturn at  $\sim 24$  K. To understand the origin of these anomalies, we prepared cesium-deficient phases,  $Cs_{1-x}P_8W_8O_{40}$ , and alkali-metal- and molybdenum-substituted phases,  $Cs_xA_yP_8W_8O_{40}$  ( $x + y > 1$ ) and  $CsP_8W_{8-x}Mo_xO_{40}$ , and measured their electrical resistivities and magnetic susceptibilities. Those results are interpreted on the basis of the tight-binding band electronic structures calculated for the  $W_4O_{18}$  chain and the  $P_8W_8O_{40}$  lattice.

The phosphate tungsten bronze  $CsP_8W_8O_{40}$  contains  $W_4O_{18}$  chains (1) made of corner-sharing  $WO_6$  octahedra.<sup>1</sup> Those chains

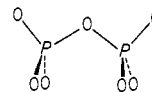


1a



1b

are linked together by  $P_2O_7$  groups (2) to form octagonal channels, where Cs atoms reside. The  $W_4O_{18}$  chains are isolated in  $Cs-$



2a



2b

$P_8W_8O_{40}$ , so this bronze is expected to be quasi-one-dimensional in electronic properties. Indeed, the electrical transport and magnetic properties of  $CsP_8W_8O_{40}$  are consistent with this picture.<sup>2</sup>

The resistivity versus temperature plot of  $CsP_8W_8O_{40}$  shows the presence of two resistivity anomalies at  $\sim 160$  and  $\sim 24$  K: it is semiconducting from  $\sim 760$  to  $\sim 160$  K, metallic between  $\sim 160$  and  $\sim 24$  K, and semiconducting below  $\sim 24$  K. In order to gain insight into the nature of the resistivity hump at  $\sim 160$

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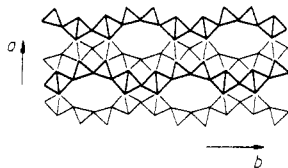
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**Table I.** Parameters and Exponents Used in the Calculations

atom	orbital	$H_{ii}$ , eV	$\zeta_1$	$\zeta_2$	$c_1^a$	$c_2^a$
O	2s	-32.3	2.275			
	2p	-14.8	2.275			
P	3s	-20.2	1.88			
	3p	-12.5	1.63			
W	5d	-10.37	4.982	2.068	0.6685	0.5424

<sup>a</sup> Contraction coefficients used in the double- $\zeta$  expansion.

**Figure 1.** Projection view of the crystal structure of  $\text{CsP}_8\text{W}_8\text{O}_{40}$ . Cesium atoms (not shown) are present in the octagonal tunnels.

K and the resistivity upturn at  $\sim 24$  K, we prepared single crystals of the cesium-deficient bronze  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  ( $x \approx 0.0-0.46$ ), the partially substituted rubidium and sodium bronzes  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $\text{A} = \text{Rb}, \text{Na}$ ), and the molybdenum-substituted bronze  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  ( $x \approx 0.1-2.1$ ) and measured their physical properties. These results are then interpreted on the basis of band electronic structure calculations on  $\text{CsP}_8\text{W}_8\text{O}_{40}$ .

### Experimental Section

Single crystals of cesium-deficient  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  ( $x \approx 0.1-0.5$ ) and partially substituted  $\text{Cs}_x\text{Rb}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $x + y > 1$ ),  $\text{Cs}_{0.89}\text{Na}_{0.17}\text{P}_8\text{W}_8\text{O}_{40}$ , and  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  ( $x \approx 0.1-2.1$ ) bronzes were grown by a solid-state reaction, similar to that described in a recent paper.<sup>2</sup> Appropriate amounts of reagent-grade  $\text{Rb}_2\text{CO}_3$  and  $\text{Na}_2\text{CO}_3$  were used in the reaction mixture for the substitution of Cs by Rb and Na, respectively. Single crystals of  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  were obtained by adding appropriate amounts of Mo metal to decomposition products of a stoichiometric mixture of  $\text{Cs}_2\text{CO}_3$ ,  $(\text{NH}_4)_2\text{HPO}_4$ ,  $\text{WO}_3$ , and W metal.

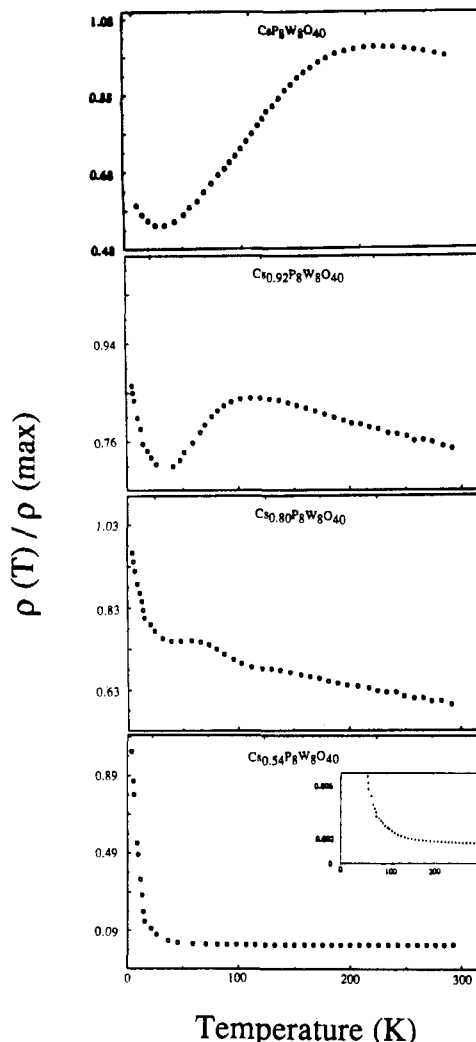
The crystals of  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$ ,  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $\text{A} = \text{Rb}, \text{Na}$ ), and  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  are golden with a metallic sheen; the crystals form in highly faceted bulky prisms. Samples selected for study had typical dimensions  $(2-4) \times 1 \times 0.5 \text{ mm}^3$ . Cell parameters were obtained by least-squares analysis of powder data obtained with a Scintag PAD V automated diffractometer employing monochromatic  $\text{Cu K}\alpha$  radiation. Silicon was used as an internal standard. The orientation of selected crystals was determined by precession photography.

Electrical resistivity was measured from 2 K to room temperature by using a standard four-probe technique with ultrasonically soldered indium contacts on oriented single crystals. The magnetic susceptibility was measured on a Quantum Design SQUID magnetometer from 2 to 270 K. The Cs stoichiometry in the Cs-deficient crystals was determined on a JEOL 4000 Superprobe electron microprobe with silver-painted electrodes; the uncertainty of Cs content determined is less than 2% by weight. Elemental analyses were obtained on a Beckman plasma emission spectrometer.

Tight-binding band electronic structure calculations<sup>3</sup> were performed on the  $\text{W}_4\text{O}_{18}$  chain (1) and also on the  $\text{P}_8\text{W}_8\text{O}_{40}$  lattice within the framework of the extended Hückel method.<sup>4</sup> The atomic parameters employed in our work are summarized in Table I.

### Crystal Structure

Diagram 1a shows a perspective view of the  $\text{W}_4\text{O}_{18}$  chain, the projection view of which is given by 1b.  $\text{P}_2\text{O}_7^{4-}$  ions (2a) link those chains by sharing six of the seven oxygen atoms. When each  $\text{P}_2\text{O}_7^{4-}$  ion is represented by 2b, the crystal structure of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  can be shown as in Figure 1. The  $\text{W}_4\text{O}_{18}$  chain is similar in structure to the  $\text{Mo}_4\text{O}_{18}$  chain found in  $\text{Li}_{0.9}\text{Mo}_6\text{O}_{17}$ <sup>5</sup> but is slightly more tilted due to its linking with the  $\text{PO}_4$  tetrahedra of  $\text{P}_2\text{O}_7^{4-}$  ions. With the formal oxidation states  $\text{Cs}^+$ ,  $\text{O}^{2-}$ , and  $\text{P}_2\text{O}_7^{4-}$ , the

**Figure 2.** Relative resistivity ( $\rho(T)/\rho(\text{max})$ ) along the  $c$  axis as a function of temperature for single crystals of  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$ .**Table II.** Lattice Parameters of Cesium-Deficient  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  Phases

sample	$a$ , Å	$b$ , Å	$c$ , Å	$V$ , Å <sup>3</sup>
$\text{CsP}_8\text{W}_8\text{O}_{40}$	13.08 (1)	12.33 (1)	5.300 (2)	855.1 (0.4)
$\text{Cs}_{0.92}\text{P}_8\text{W}_8\text{O}_{40}$	13.07 (1)	12.33 (1)	5.298 (5)	854.1 (0.4)
$\text{Cs}_{0.80}\text{P}_8\text{W}_8\text{O}_{40}$	13.06 (1)	12.34 (1)	5.301 (2)	854.0 (0.6)
$\text{Cs}_{0.54}\text{P}_8\text{W}_8\text{O}_{40}$	13.06 (1)	12.33 (1)	5.296 (1)	852.8 (0.3)

formal oxidation state of W in  $\text{CsP}_8\text{W}_8\text{O}_{40}$  is 4.875. Thus, the electron counting on W is  $d^{1.125}$ , and there are 4.5 electrons to fill the d-block bands of the  $\text{W}_4\text{O}_{18}$  chain per formula unit ( $\text{W}_4\text{O}_{18}$ ).

### Physical Properties

**A.  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  ( $x \approx 0.0-0.46$ ).** The electrical resistivities of  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  are summarized in Figure 2. The broad resistivity peak of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  at  $\sim 160$  K is gradually shifted toward lower temperatures, as the Cs content is decreased. At  $x = 0.20$  this peak is barely observable, and at  $x = 0.46$  only semiconducting behavior is seen in the whole temperature range. The resistivity upturn of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  at  $\sim 24$  K does not seem to be affected significantly by decreasing Cs concentration. Qualitative Seebeck measurements on  $\text{CsP}_8\text{W}_8\text{O}_{40}$  showed p-type conductivity at room temperature but n-type conductivity below 100 K.<sup>2</sup>

The temperature variation of the magnetic susceptibility of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  is Pauli paramagnetic,<sup>2</sup> indicative of metallic behavior, while that of  $\text{Cs}_{0.54}\text{P}_8\text{W}_8\text{O}_{40}$  shows paramagnetic behavior found for systems with localized electrons (see Figure 3a).

As shown in Table II, the cell parameters of the Cs-deficient bronzes with decreasing Cs content do not change significantly

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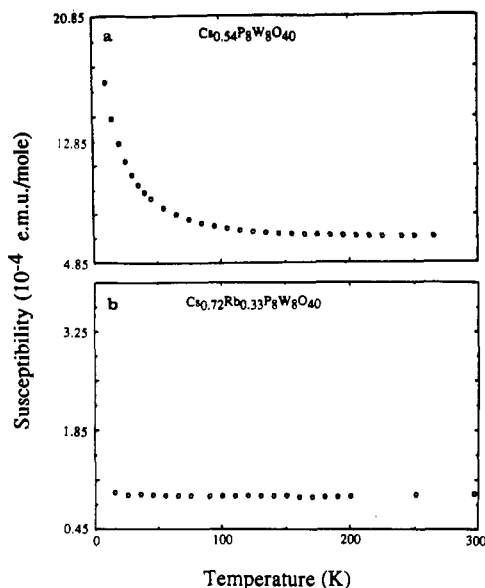


Figure 3. Magnetic susceptibility as a function of temperature for single crystals of (a)  $\text{Cs}_{0.54}\text{P}_8\text{W}_8\text{O}_{40}$  and (b)  $\text{Cs}_{0.72}\text{Rb}_{0.33}\text{P}_8\text{W}_8\text{O}_{40}$ .

Table III. Lattice Parameters of  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  (A = Rb, Na) Phases

sample	a, Å	b, Å	c, Å	V, Å <sup>3</sup>
$\text{CsP}_8\text{W}_8\text{O}_{40}$	13.08 (1)	12.33 (1)	5.300 (2)	855.1 (0.4)
$\text{Cs}_{0.97}\text{Rb}_{0.16}\text{P}_8\text{W}_8\text{O}_{40}$	13.07 (1)	12.33 (1)	5.303 (2)	854.8 (0.4)
$\text{Cs}_{0.89}\text{Rb}_{0.21}\text{P}_8\text{W}_8\text{O}_{40}$	13.06 (1)	12.33 (1)	5.305 (3)	854.1 (0.6)
$\text{Cs}_{0.72}\text{Rb}_{0.33}\text{P}_8\text{W}_8\text{O}_{40}$	13.06 (2)	12.36 (1)	5.301 (4)	855.3 (1.1)
$\text{Cs}_{0.89}\text{Na}_{0.17}\text{P}_8\text{W}_8\text{O}_{40}$	13.08 (2)	12.34 (1)	5.306 (5)	856.1 (0.8)

compared to those of the host,  $\text{CsP}_8\text{W}_8\text{O}_{40}$ . This suggests that the observed changes in the broad resistivity peak at  $\sim 160$  K are electronic rather than structural in origin. The observed systematic decrease in the unit cell volume is consistent with decreasing amounts of Cs in the cavities.

**B.  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  (A = Rb, Na).**  $\text{Cs}_x\text{Rb}_y\text{P}_8\text{W}_8\text{O}_{40}$  compounds with  $x + y > 1$ , as determined by chemical analysis, are obtained as a result of attempts to prepare  $\text{Cs}_{1-x}\text{Rb}_x\text{P}_8\text{W}_8\text{O}_{40}$ . The absolute resistivity of  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  is smaller along the *c* axis than in the *ab* plane, thereby showing a pseudo-1D character. Figure 4 shows the resistivity determined for oriented single crystals of  $\text{Cs}_{0.89}\text{Na}_{0.17}\text{P}_8\text{W}_8\text{O}_{40}$ ,  $\text{Cs}_{0.97}\text{Rb}_{0.16}\text{P}_8\text{W}_8\text{O}_{40}$ ,  $\text{Cs}_{0.89}\text{Rb}_{0.21}\text{P}_8\text{W}_8\text{O}_{40}$ , and  $\text{Cs}_{0.72}\text{Rb}_{0.33}\text{P}_8\text{W}_8\text{O}_{40}$ . The resistivity hump at  $\sim 160$  K is absent for measurements along the crystallographic *c* axis in all the samples. However, a semiconductor-to-metal transition is seen in the resistivity of  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  when measured in the *ab* plane, although the transition temperature,  $T_{sm}$ , does not scale with  $x + y$  and appears to be sample dependent. Rb substitution broadens the metal-insulator transition at  $\sim 24$  K. This broadening increases with the Rb concentration, regardless of the  $x + y$  value. Thus, this may be attributed to the increasing concentration of defects, as observed for the blue bronze  $(\text{K}_{1-x}\text{Rb}_x)_{0.3}\text{MoO}_3$ <sup>6</sup> and the purple bronze  $(\text{Li}_{1-x}\text{Na}_x)_{0.9}\text{Mo}_6\text{O}_{17}$ .<sup>7</sup>

The magnetic susceptibilities for all  $\text{Cs}_x\text{Rb}_y\text{P}_8\text{W}_8\text{O}_{40}$  phases show temperature-independent (i.e. Pauli paramagnetic) behavior, indicative of metallic character. As a representative example, Figure 3b shows the magnetic susceptibility for  $\text{Cs}_{0.72}\text{Rb}_{0.33}\text{P}_8\text{W}_8\text{O}_{40}$ . The unit cell parameters of the Na- and Rb-substituted  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  phases are similar to those of the host (Table III).

**C.  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  ( $x \approx 0.1$ – $2.1$ ).** Figure 5 shows the electrical resistivity data of the molybdenum-substituted bronzes  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$ . This isoelectronic substitution gradually suppresses the metallic region of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  (Figure 2). As the

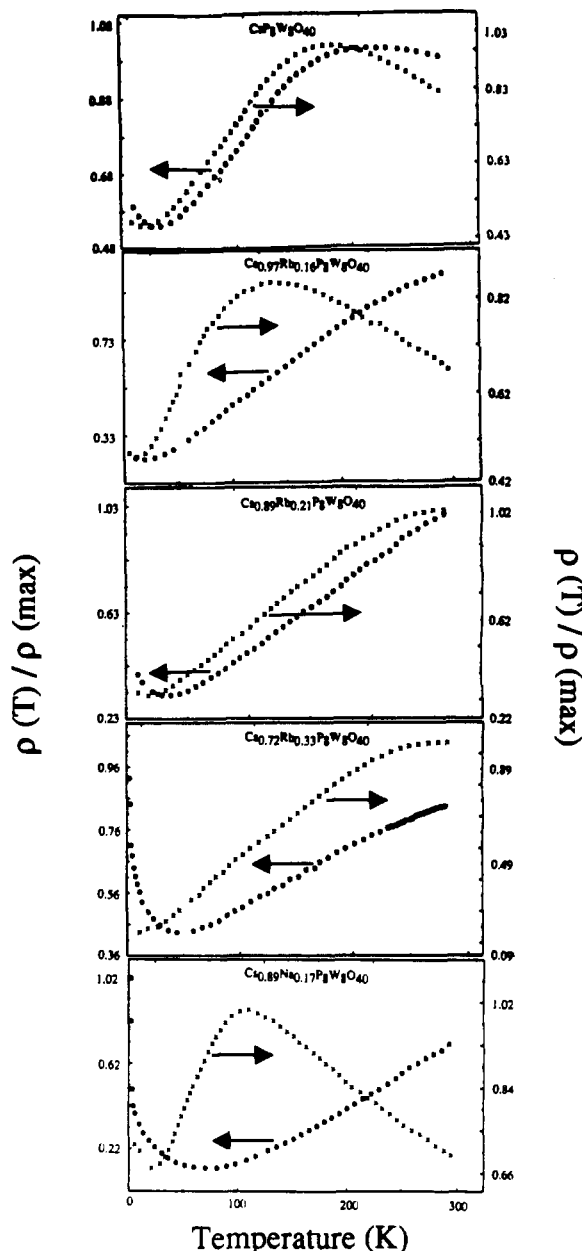


Figure 4. Relative resistivity ( $\rho(T)/\rho(\max)$ ) along the *c* axis (O) and the *ab* plane (X) as a function of temperature in single crystals of  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$ .

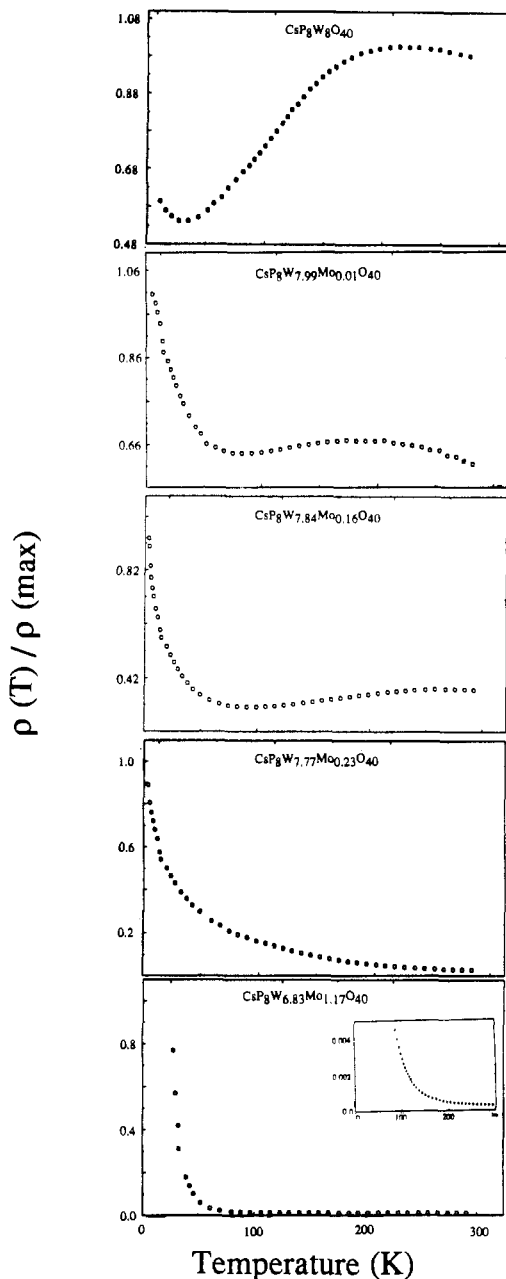
Table IV. Lattice Parameters of Mo-Substituted  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  Phases

sample	a, Å	b, Å	c, Å	V, Å <sup>3</sup>
$\text{CsP}_8\text{W}_8\text{O}_{40}$	13.08 (1)	12.33 (1)	5.300 (2)	855.1 (0.4)
$\text{CsP}_8\text{W}_{7.99}\text{Mo}_{0.01}\text{O}_{40}$	13.10 (1)	12.34 (1)	5.298 (3)	856.5 (0.3)
$\text{CsP}_8\text{W}_{7.84}\text{Mo}_{0.16}\text{O}_{40}$	13.09 (1)	12.33 (1)	5.300 (6)	855.4 (0.5)
$\text{CsP}_8\text{W}_{7.77}\text{Mo}_{0.23}\text{O}_{40}$	13.07 (1)	12.33 (1)	5.306 (3)	855.1 (0.3)
$\text{CsP}_8\text{W}_{6.83}\text{Mo}_{1.17}\text{O}_{40}$	13.07 (1)	12.33 (1)	5.296 (2)	853.4 (0.5)
$\text{CsP}_8\text{W}_{5.9}\text{Mo}_{2.1}\text{O}_{40}$	13.06 (1)	12.30 (1)	5.291 (5)	850.0 (0.9)

amount of Mo substitution increases to  $x = 0.23$ , the resulting bronze shows semiconducting behavior. The temperature dependence of the magnetic susceptibility for this sample shows paramagnetic behavior consistent with the resistivity data.

The unit cell dimensions of the  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$  bronzes (Table IV) do not differ much from those of the host  $\text{CsP}_8\text{W}_8\text{O}_{40}$  since the effective radius of Mo is comparable with that of W. However, there seems to be a decrease in the cell volume for substantial Mo substitution. Attempts to synthesize the Mo analogue  $\text{CsP}_8\text{Mo}_3\text{O}_{40}$  resulted in unidentified multiple phases. Cs–P–Mo–O bronzes could probably be synthesized, but the composition would have to be altered. This possibility was confirmed by the recent report

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 (7) Ramanujachary, K. V.; Collins, B. R.; Greenblatt, M.; McNally, P.; McCarroll, W. H. *Solid State Ionics* 1986, 22, 105.



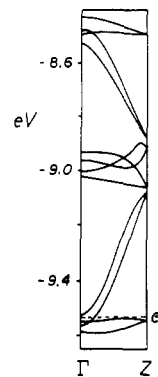
**Figure 5.** Resistivity along the  $c$  axis as a function of temperature for single crystals of  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$ .

on the preparation of cesium molybdenum phosphate cluster compounds  $\text{Cs}_3\text{Mo}_5\text{P}_6\text{O}_{25}$ <sup>8</sup> and  $\text{Cs}_4\text{Mo}_8\text{P}_{12}\text{O}_{52}$ .<sup>9</sup>

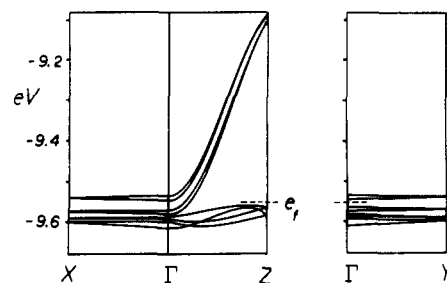
#### Band Electronic Structure

The  $t_{2g}$ -block bands of the  $\text{W}_4\text{O}_{18}$  chain 1 are shown in Figure 6, where the dashed line represents the Fermi level appropriate for 4.5 d electrons per unit cell. These bands are essentially identical in nature with those of the ideal  $\text{Mo}_4\text{O}_{18}$  chain described in detail elsewhere in connection with  $\text{Li}_{10}\text{Mo}_6\text{O}_{17}$ .<sup>10</sup> Figure 7 shows the  $t_{2g}$ -block bands (only the bottom portion) of the  $\text{P}_8\text{W}_8\text{O}_{40}^-$  lattice. Since this lattice has two  $\text{W}_4\text{O}_{18}$  chains per unit cell, it has twice as many  $t_{2g}$ -block bands as does the  $\text{W}_4\text{O}_{18}$  chain and nine electrons to fill them. Except for the band doubling and a very small splitting in each pair of bands, the  $t_{2g}$ -block bands of the  $\text{P}_8\text{W}_8\text{O}_{40}^-$  lattice are identical with those of the  $\text{W}_4\text{O}_{18}$  chain.

The most important features of the  $t_{2g}$ -block bands in Figures 6 and 7 can be summarized as follows: (a) The Fermi level lies

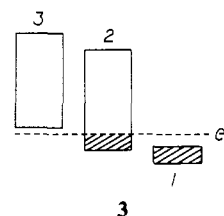


**Figure 6.** Dispersion relations of the  $t_{2g}$ -block bands of the  $\text{W}_4\text{O}_{18}$  chain, where  $\Gamma = 0$  and  $Z = c^*/2$ . The Fermi level (dashed line) is for 4.5 d electrons per formula unit  $\text{W}_4\text{O}_{18}$ .



**Figure 7.** Dispersion relations of the bottom portion of the  $t_{2g}$ -block bands calculated for the  $\text{P}_8\text{W}_8\text{O}_{40}^-$  lattice, where  $\Gamma = (0, 0, 0)$ ,  $X = (a^*/2, 0, 0)$ ,  $Y = (0, b^*/2, 0)$ , and  $Z = (0, 0, c^*/2)$ . The Fermi level is for nine d electrons per unit cell.

close to, but above, the top of the flat filled bands. (b) The Fermi level lies close to, but below, the bottom of the dispersive, one-dimensional, empty bands. (c) There exist one-fourth-filled one-dimensional bands, which are dispersive. These three characteristics are schematically represented by diagram 3, where



bands 2 and 3 refer to the dispersive one-dimensional bands while band 1 refers to the flat bands. In the following, we discuss the electrical properties of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  and its substituted analogues on the basis of diagram 3.

#### Correlation of Electrical Properties with Band Electronic Structure

**A. Resistivity Anomalies of  $\text{CsP}_8\text{W}_8\text{O}_{40}$ .** In understanding the nature of the  $\sim 160$  K resistivity hump in  $\text{CsP}_8\text{W}_8\text{O}_{40}$  (Figure 2), it is necessary to recall that the carriers are holes above 160 K but electrons in the 24–160 K region.<sup>2</sup> In diagram 3 the Fermi level is close to the top of band 1 and also to the bottom of band 3. Thus, with thermal energy, electrons are excited from band 1 to bands 2 and 3. One-dimensional bands have a high density of states at their bottom (and at their top as well). The bottom of band 2 is already filled, while that of band 3 is empty, so that thermal electron excitation into the bottom of band 3 either from band 2 or from band 1 is quite likely.<sup>11</sup> This creates holes in band 1. The hole contribution to conductivity from band 1 is activated since the effective mass of this band is large due to its narrowness. If electrons of band 2 and 3 do not significantly contribute to

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conductivity due to thermal motions of the lattice, the hole contribution will dominate. As the temperature is lowered below 160 K, deexcitation of electrons from bands 2 and 3 to band 1 will remove the holes of band 1. Further, thermal motions of the lattice are reduced. Thus, electrons of band 2 contribute significantly to conductivity, so that the  $\text{CsP}_8\text{W}_8\text{O}_{40}$  bronze is metallic and its carriers are electrons below 160 K. The resistivity upturn at  $\sim 24$  K can be ascribed to a charge density wave (CDW) associated with the one-fourth filled band 2, since it is a strongly one-dimensional one. It should be possible to observe a CDW with vector  $q_{\text{CDW}} \approx 0.25c^*$ .

**B.  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$ .** Figure 2 shows that the resistivity hump of  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  (at  $\sim 160$  K for  $x = 0$ ) is gradually shifted to a lower temperature as  $x$  increases toward 0.5 and that, at  $x \approx 0.5$ , the bronze becomes semiconducting at all temperatures. These observations can be explained as follows: as  $x$  increases from 0.0 to 0.5, the number of electrons in band 2 decreases from 0.5 to 0.25. As the electron density in band 2 decreases, screening among the electrons of the band is reduced, so that the electrons become more susceptible toward localization around  $\text{Cs}^+$ -rich neighbors. Therefore, even electrons of band 2 may require activated conductivity. Thus, the resistivity hump of  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  is gradually shifted to a lower temperature as the Cs content is decreased, and beyond a certain value of  $x$ , the bronze becomes semiconducting.

**C.  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $\text{A} = \text{Rb}, \text{Na}$ ).** Figure 4 shows that, in contrast to  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$ , the  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  bronze does not have a resistivity hump when measured along the  $c$  crystallographic direction. For the partially substituted bronze,  $x + y > 1$ , so that it has more  $d$  electrons than does  $\text{CsP}_8\text{W}_8\text{O}_{40}$ . In diagram 3, which is applicable for  $x + y = 1$ , thermal excitation probability is large for the excitation to the bottom of band 3, as already mentioned. Thus, provided that the  $x + y$  value of the Rb- and Na-substituted bronze is large enough to fill the bottom portion of band 3, thermal excitation from band 1 would be weak. Then the hole contribution to conductivity from band 1 is diminished so that electron contribution from band 2 becomes dominant. As a result, the re-

sistivity hump along the  $c$  axis found for  $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  will be absent in  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $x + y > 1$ ). The sample-dependent semiconductor-to-metal transition seen in the resistivity of  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  samples in the  $ab$  plane cannot be accounted for in terms of the band diagram, and its origin is not clear.

**D.  $\text{CsP}_8\text{W}_{8-x}\text{Mo}_x\text{O}_{40}$ .** According to Figure 5, the metallic character of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  quickly disappears when a small amount of W is substituted with Mo. This result is expected since the random distribution of Mo atoms in the  $\text{W}_4\text{O}_{18}$  chains will disrupt the band electron contribution to conductivity.

#### Concluding Remarks

$\text{CsP}_8\text{W}_8\text{O}_{40}$  exhibits a broad resistivity hump at  $\sim 160$  K and a resistivity upturn at  $\sim 24$  K. To examine the origin of these resistivity anomalies, we prepared its electron-deficient and electron-rich analogues ( $\text{Cs}_{1-x}\text{P}_8\text{W}_8\text{O}_{40}$  and  $\text{Cs}_x\text{A}_y\text{P}_8\text{W}_8\text{O}_{40}$  ( $x + y > 1$ ), respectively), measured their physical properties, and performed tight-binding band electronic structure calculations on the  $\text{P}_8\text{W}_8\text{O}_{40}^-$  lattice. The present work suggests the  $\sim 160$  K resistivity hump to be due to the fact that the Fermi level of the  $\text{P}_8\text{W}_8\text{O}_{40}^-$  lattice, which occurs near the bottom of dispersive one-dimensional bands, lies very close to the top of flat, filled bands and also very close to the bottom of dispersive, one-dimensional, empty bands. Our study also suggests that the resistivity upturn of Cs at  $\sim 24$  K stems from a CDW associated with the dispersive one-dimensional bands.

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## Structural and Electronic Origin of the Three-Dimensional Electrical Properties of $\text{P}_8\text{W}_{12}\text{O}_{52}$ and Its Inserted and Substituted Analogues $\text{A}_x\text{P}_8\text{W}_{12}\text{O}_{52}$ ( $\text{A} = \text{Li}, \text{Na}$ ) and $\text{P}_8\text{W}_{12-x}\text{Mo}_x\text{O}_{52}$

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Single crystals of  $\text{P}_8\text{W}_{12}\text{O}_{52}$  and its inserted and substituted analogues  $\text{A}_x\text{P}_8\text{W}_{12}\text{O}_{52}$  ( $\text{A} = \text{Li}$ ,  $x = 0.16$ ;  $\text{A} = \text{Na}$ ,  $x = 0.22$ ) and  $\text{P}_8\text{W}_{12-x}\text{Mo}_x\text{O}_{52}$  ( $x = 0.28, 0.40, 0.68$ ) were prepared, and their electrical resistivities and magnetic susceptibilities were measured. All of these bronzes are found to be isotropic metals. Our tight-binding band electronic structure calculations on  $\text{P}_8\text{W}_{12}\text{O}_{52}$  show that the three-dimensional metallic character originates from a three-dimensional linking of  $\text{WO}_6$  octahedra in the lattice and the relatively high oxidation state of W. Examination of the structural relationship between  $\text{P}_8\text{W}_{12}\text{O}_{52}$  and  $\text{CsP}_8\text{W}_8\text{O}_{40}$  led to a new synthetic route for the preparation of  $\text{CsP}_8\text{W}_8\text{O}_{40}$  from  $\text{P}_8\text{W}_{12}\text{O}_{52}$ . With use of this new method,  $\text{Ti}_2\text{P}_8\text{W}_8\text{O}_{40}$  (isostructural with  $\text{CsP}_8\text{W}_8\text{O}_{40}$ ) was synthesized for the first time.

#### Introduction

Tungsten bronzes of the diphosphate ions  $\text{P}_2\text{O}_7^{4-}$  have open framework channel structures, in which alkali-metal and other metal ions can be intercalated. So far, three types of channel

structures have been identified in diphosphate tungsten bronzes:  $\text{CsP}_8\text{W}_8\text{O}_{40}$  has octagonal channels,<sup>1</sup> a family of  $\text{A}_x(\text{P}_2\text{O}_7)_4(\text{WO}_3)_{4m}$  ( $\text{A} = \text{K}$  with  $m = 5-11$ ;<sup>2</sup>  $\text{A} = \text{Rb}$  with  $m = 4-11$ ;<sup>3</sup>  $\text{A} = \text{Ti}$  with  $m = 8$ ;<sup>4</sup>  $\text{A} = \text{Ba}$  with  $m = 6-10$ ) has hexagonal

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